Self-sustained Waves of Exothermic Dissolution in Reactive Multilayer Nanofoils

The Phenomenon of Wave Localization for Solid State Self-propagating Reactions

Present-day name:



Self-propagating High-temperature Synthesis (SHS) A.G. Merzhanov, V.M. Shkiro, I.P. Borovinskaya, 1967

$$Ta + C \rightarrow \frac{1}{2} Ta_2C + \frac{1}{2} C + Q_1 \rightarrow TaC + Q_2$$

 $T_{ad} = 2730 \text{ K}, T_m = 3100 \text{ K}$



lu

Combustion

products (solid)

Combustion zone











ignition

front propagation

cooling

Solid Flame:

Recent Advances

Ta + C Reactive Media after High Energy Ball Milling (HEBM) (2012)





Comparison of XRD patterns for Ta-C system before (a) and after (b) HEBM

Reaction Front Propagation in Ta-C Nano Heterogeneous System



Ignition temperature of each composite particles ~450°C !!!

Composite Reactive Nano-particles by HEBM: *Definition*



Particles obtained by intensive short term High Energy Ball Milling (HEBM) of reactive powder mixtures.

Bright-field TEM image of Ni-Al Media: Nano Mixing



Bright-field TEM image: Atomic Mixing







Reactive Nano-Foils





Production of Reactive Multilayer Nanofoils



T.Weihs et al. – J.Hopkins Univer., USA, 1996; P.Tsygankov et al. – Bauman Technical University, Russia , 2000.

Reactive Multilayer Nanofoil





Quenching of the Combustion Front





Microstructure evolution behind the reaction front



Grain size Growth in Quenched Front



Dynamics of grain size (1) and bilayer thickness (2) behind the reaction front.

Formation of solid NiAl grains after saturation of the Al-Ni melt



Phase Formation in Reaction Front

<u>50 nm</u>

d _{exp} (A)	d _{database} (A)			
	Ni	Al	AlNi	Al3Ni
4.02				4.03 <011>
2.87			2.86 <100>	
2.03	2.03 <111>	2.03 <200>	2.03 <110>	2.01 <022>
1.80	1.76 <200>			1.82 <231>
1.64			1.65 <111>	1.66 <400>
1.43		1.43 <220>	1.43 <200>	1.43 <142>
1.35				1.35 <250>
1.26	1.25 <220>	1.22 <311>	1.28 <210>	1.26 <511>
1.17		1.17 <222>		1.17 <114>
1.08	1.06 <311>			1.08 <053>
1.02	1.02 <222>	1.01 <400>		1.02 <503>

EDS Analyses

Concentration profiles across the layers in the initial foil (a) and reaction zone (b) in the sample with quenched reaction wave (TEM+EDS at thin cross-sections).

General Reaction Scheme

(A) Photo of the exothermic wave (A);

- (B) SEM picture of the quenched reaction front;
- (C) TEM pictures of the reaction zone;
- (D) dynamics of the thickness of Ni layer (Δ),
 - Al-rich layer (\Box) , and bilayer period (\circ) , I-initial structure,
 - II Ni dissolution,
 - III NiAl precipitation;
- (E) schematic draw of the wave structure.

The sequence of the process stages can be represented as follows:

 $\begin{array}{l} \mathsf{Ni}(s) + \mathsf{AI}(s) \ \ \ \ \mathsf{B} \ \mathsf{Ni}(s) + \mathsf{AI}(m);\\ \mathsf{Ni}(s) + \mathsf{AI}(m) \ \ \ \ \ \mathsf{B} \ \mathsf{Ni}(s) + \mathsf{AINi}_{\mathsf{x}}(m);\\ \mathsf{Ni}(s) + \mathsf{AINi}_{\mathsf{x}}(m) \ \ \ \ \ \mathsf{B} \ \mathsf{AINi}(s), \end{array}$

At the first stage, the aluminum melts just above the reaction front due to heat flow from the reaction zone.

Melting of Al initiates partial dissolution of the

Ni layers (stage 2) into Al melt until the saturation concentration ($x \sim 0.25 - 0.3$) is reached.

At the third stage, solid grains of the NiAl B2-phase nucleate and grow.

The characteristic feature of the process is that the grains have different crystallographic orientations and are separated by liquid/amorphous interlayers. The existence of the liquid between the solid grains allows Ni to dissolve into the melt by means of liquid-state diffusion, which is much faster than solid state diffusion across the intermetallic solid compound.

